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Supercapacitors based on two dimensional VO₂ nanosheet electrodes in organic gel electrolyte



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ABSTRACT

 VO_2 is a low band-gap semiconductor with relatively high conductivity among transition metal oxides, which makes it an interesting material for supercapacitor electrode applications. The performance of VO_2 as supercapacitor electrode in organic electrolytes has never been reported before. Herein, two-dimensional nanosheets of VO_2 are prepared by the simultaneous solution reduction and exfoliation from bulk V_2O_5 powder by hydrothermal method. A specific capacitance of 405 Fg^{-1} is achieved for VO_2 based supercapacitor in an organic electrolyte, in three electrode configuration. The symmetric capacitor based on VO_2 nanosheet electrodes and the liquid organic electrolyte exhibits an energy density of 46 Wh kg^{-1} at a power density of 1.4 kW kg^{-1} at a constant current density of 1 Ag^{-1} . Furthermore, flexible solid-state supercapacitors are fabricated using same electrode material and Alumina-silica based gel electrolyte. The solid-state device delivers a specific capacitance of 145 Fg^{-1} and a device capacitance of 36 Fg^{-1} at a discharge current density of 1 Ag^{-1} . Series combination of three solid state capacitors is capable of lighting up a red LED for more than 1 minute.

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1. Introduction

Inorganic graphene-like 2D materials with single or few layers have received considerable research interest recently [1–4], as these materials show distinctive physical and electrical properties, which makes them potentially useful for a variety of new applications [5–7]. One of the unique properties of ultrathin inorganic 2D nanosheets is their large surface area and their ability to undergo redox reactions efficiently, which makes them candidates for fabricating high energy density storage devices [8,9]. Supercapacitors are high power devices that can store and deliver energy much quicker than the batteries, but they do have lower in energy densities [10-12]. To increase the energy density of supercapacitors different nanostructured materials have been investigated including redox polymers and oxides [13-21]. Although, higher energy densities have been achieved with nanostructured materials, this has often been at the expense of modest power density and cycling stability. The approach to improving the energy storage performance of pseudocapacitors should aim at reducing the ion diffusion path length and increasing

http://dx.doi.org/10.1016/j.electacta.2016.10.109 0013-4686/© 2016 Elsevier Ltd. All rights reserved. the electronic conductivity as well as the surface area of the active material. Since the ionic diffusion in 2D materials is high [22], developing a thin 2D inorganic material could result in high energy density and power density supercapacitors. The 2D metal oxides such as V₂O₅ and MoO₃ etc. have been utilized as supercapacitor electrode materials [23-48]. Among all these, vanadium oxide is gaining considerable research interest due to its ability to exist in variable oxidation states (e.g., V_2O_5 , V_2O_3 , and V_4O_7) [38–49]. The electronic structure, charge density, transport properties, oxidation state and phase transitions of different vanadium oxides are expected to vary with the composition. Among the oxides of vanadium, V₂O₅ has been quite extensively investigated for energy storage applications [50-52]. On the other hand, VO₂ is an exciting phase having a low band gap of 0.7 eV, which changes monoclinic to tetragonal structure reversibly at about 68°C and undergoes semiconductor-to-metal transition.

There are few reports of VO₂ as a supercapacitor electrode material [53–59]. Pan et al. have reported the conductivity of VO₂ can be enhanced by treating with hydrogen and hence supercapacitor performance [54]. At a constant current density of 1 Ag^{-1} , the specific discharge capacitance and specific energy density of hydrogen treated VO₂ was stabilized at around 300 Fg⁻¹ and 17 Wh kg⁻¹ respectively, which was nearly four times higher than that of pristineVO₂. This remarkable enhancement in the

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performance was attributed to the reduction of VO₂ resistance by nearly three orders of magnitude through H₂ treatment [54]. Zhao et al. have utilized the porous hierarchical VOx@carbon composites as supercapacitor electrode materials [56]. Zhao et al. have reported the use of VO₂-reduced graphene oxide composite as an efficient electrode material for electrochemical energy storage applications [57,60]. Recently, Wang et al. prepared the hierarchical network framework graphene/VO₂ nanobelt composite hydrogels which exhibited a maximum energy density of 21.3 Wh kg⁻¹ at a constant current density of 1 Ag^{-1} [58]. However, to the best of our knowledge, no reports are available on the electrochemical performance of VO₂ in an organic electrolyte or on flexible solid state capacitors based on VO₂ with organic gel electrolyte.

In this study, we demonstrate a process for the simultaneous reduction and exfoliation of bulk V_2O_5 powder into an assembly of 2D VO_2 nanosheets by hydrothermal method. By optimizing process conditions and choosing the appropriate hydrophilic binder, we could achieve a specific capacitance of 217 Fg⁻¹ at a scan rate of 5 mV s⁻¹ for VO_2 based symmetric capacitor in a propylene carbonate (PC) based organic electrolyte. Symmetric supercapacitor displays an energy density of 46 Wh kg⁻¹ at a power density of 1.4 kW kg⁻¹ and a constant current density of 1 Ag⁻¹. The flexible solid-state supercapacitor fabricated using VO_2 electrodes and gel organic electrolyte could deliver a specific capacitance of 145 Fg⁻¹ and a gravimetric capacitance of 36 Fg⁻¹ at a discharge current density of 1 Ag⁻¹.

2. Experimental

2.1. Chemicals

Ammonium metavanadate (NH₄VO₃), Hydrogen peroxide, Oxalic acid (H₂C₂O₄), Propylene Carbonate (C₄H₆O₃), Lithium perchlorate (LiClO₄), Polytetrafluoroethylene (PTFE), Sodium Carboxymethylcellulose (CMC), ethanol (C₂H₅OH), Fumed Silica (Aldrich) were used without further purification.

2.2. Synthesis of VO_2 2D nanosheets from V_2O_5 powder

Ammonium metavanadate powder was heated to $450 \,^{\circ}$ C in air for two hours in a tube furnace at a heating rate of $10 \,^{\circ}$ C per minute to obtain orange-red color V₂O₅ powder. The 0.25 g of V₂O₅ powder was dissolved in 20 ml of water using 30% H₂O₂ and stirred for 30 minutes. To this, 1 g of oxalic acid was added and stirred for additional 30 minutes. The resulting dark orange-red colored solution was transferred into Teflon hydrothermal reaction autoclaves and heated to 180 °C for 24 hours. The resulting greenish black colored gel was centrifuged and washed several times with water followed by ethanol and dried at room temperature.

2.3. Instrumentation

Morphology of VO₂ was characterized by field emission scanning electron microscope (FE-SEM, FEI, Nova Nano) and transmission electron microscope (TEM, JEOL 2100F). The X-ray diffraction (XRD) was performed using Bruker diffractometer (CuK α), Brunauer-Emmett-Teller (BET) surface area measurement was performed by using NOVA 3200e system. The VO₂ sample was degassed at 373 K under vacuum for 12 h before BET surface area measurements.

Classical three-electrode configuration was used for the electrochemical characterization of VO₂ nanosheets. VO₂ modified carbon paper was used as a working electrode, Ag/Ag⁺ as a reference electrode and Pt wire as a counter electrode. Capacitive performance of the VO₂ nanosheets were conducted in symmetric two electrode configuration. Coin type device was fabricated in symmetric cell configuration using two VO₂ electrodes of identical masses (1 mg each) with 1 M LiClO₄ in propylene carbonate as a liquid electrolyte and a separator.

2.4. Electrochemical Characterization

The electrodes were prepared by mixing the 85% of VO₂ with 10% weight of acetylene black and 5% binder CMC in 20 ml of water and sonicated for 30 minutes to obtain a stable dispersion solution. The dispersed solution was drop cast onto the carbon paper and dried in the oven at 50 °c for 12 hours.

The gel electrolyte was prepared by dissolving the fumed silica in a minimum quantity of propylene carbonate solution containing 1 M LiClO₄. The solvent was allowed to evaporate in the air until it became a gel. The resultant gel is transparent and stable and used as a gel electrolyte in solid-state device fabrication.

The electrochemical properties and capacitance measurements of the VO₂ electrodes were studied by cyclic voltammetry (CV) and galvanostatic charge-discharge cycling at different current densities. The specific capacitance C_{sp} was calculated from the charge discharge curves as according to the following Eqs. (1) and (2) respectively for three and two electrode configurations

$$C_{sp} = \frac{1}{m(\frac{dV}{dt})} \tag{1}$$

$$C_{sp} = \frac{2 I}{m(\frac{dV}{dt})} \tag{2}$$

Gravimetric cell capacitance of the flexible solid state supercapacitor device was calculated using the following equation

$$C_g = \frac{1}{M(\frac{dV}{dt})} \tag{3}$$



Fig.1. FESEM images of (a) VO₂ nanosheets, (b) higher magnification image, (c) electrode composite of VO₂, carbon and CMC. Inset of (c) shows the optical image of VO₂ with CMC dispersion solutions along with carbon in water, optical image was taken after 45 minutes of their preparation.



Fig. 2. (a) TEM images showing the nanosheet morphology (b) Higher resolution image showing the lattice fringes, (c) STEM image of nanosheets and (d-f) elemental mapping of vanadium, oxygen and carbon.

Where *I* is the current density during discharge process, *dt* is the discharge time, *dV* is the potential window and m (= 1 mg) is the mass of active material in one electrode and M = 2m = 2 mg.

The energy (E) and power densities (P) for the supercapacitors were calculated from charge–discharge curves at different current densities using Eqs. (4) and (5) respectively.

$$E = \frac{1}{2} C_{sp} \Delta V^2 \tag{4}$$

where ' ΔV ' is the potential window of discharge process.

$$P = \frac{E}{\Delta t} \tag{5}$$

3. Results and Discussions

SEM analysis is used to investigate the morphology of VO_2 and VO_2 /binder/carbon composite. The SEM images of VO_2 exhibit twodimensional sheet-like morphology, grown in bundles, and the majority of the nanosheets are aligned vertically as shown in Fig. 1 a and b. The SEM image of the electrode composite of VO_2 , carbon and CMC binder is shown in Fig. 1c. Hydrophilic nature of CMC binder and VO_2 leads to a uniform stable dispersion of VO_2 (inset of Fig. 1c) with CMC binder and carbon in water. Details of the structural characterizations of VO_2 nanosheets are shown in Fig. S1.

TEM analysis is used for the structural analysis of the VO₂ nanosheets. TEM images of 2D VO₂ nanosheets are shown in Fig. 2a. The TEM image shows the ultrathin nanosheets with lateral dimensions ranging from 60 nm to 2 μ m. Fig. 2b shows the high-resolution TEM image displaying the lattice fringes with a spacing of 0.2 nm, which corresponds to the monoclinic VO₂ phase. The scanning transmission electron micrograph (STEM) image of the VO₂ nanosheets is shown in Fig. 2c. Fig. 2d–f represents the elemental mapping of the region selected in Fig. 2c. The images clearly show the uniform distribution of the elements V, O, and C in the nanosheet. The elements V and O originate from VO₂ and the C originates from the oxalic acid which was used to reduce the bulk powder V₂O₅ during the hydrothermal process.

Supercapacitor electrodes are prepared by drop casting of stable suspension of VO_2 nanosheets, conducting carbon and the CMC binder in water- over carbon paper, as shown in Fig. 3. The electrochemical performance of the VO_2 electrodes was



Fig. 3. Schematic of the preparation of the VO₂ nanosheet electrode with CMC binder on graphitized carbon paper.

investigated using cyclic voltammetry and galvanostatic chargedischarge cycling using three electrode configuration. The CV curves obtained at different scan rates show faradaic behavior in the potential range (-0.3 V to 1.1 V) studied (Fig. 4a). The shape remains unchanged at scan rates ranging from 2–50 mV/s with minimal peak shift, indicating efficient electron transfer in the electrodes. The galvanostatic charge-discharge curves at different current densities are shown in Fig. 4b. A specific capacitance of about 405 Fg⁻¹ is achieved for the VO₂ electrode in three electrode configuration in the organic electrolyte (1 M LiClO₄ in PC) at a constant current density of 1 Ag^{-1} . This value is nearly 1.6 times higher than the specific capacitance reported for V₂O₅ nanosheets in 1 M KCl aqueous electrolyte in our earlier work [61]. The two electrode configuration is more appropriate to study the capacitance as it mimics the design of a real capacitor. Hence, symmetric supercapacitors are constructed using VO₂ electrodes and an



Fig. 4. CVs at different scan rates of (a) VO₂/C/CMC electrode and (c) it's symmetric supercapacitor with PC/LiClO₄ electrolyte, Charge-discharge characteristics at different current densities of (b) VO₂/C/CMC electrode and (d) its symmetric supercapacitor with PC/LiClO₄ electrolyte. (e) Comparison of Ragone plot for VO₂/C/CMC based symmetric supercapacitor with literature data. (f) Cycling performance of the symmetric capacitor at a constant current density of 10 Ag⁻¹.



Fig. 5. (a) Nyquist plot and (b) Variation of specific capacitance with frequency for symmetric supercapacitor based on VO₂/C/CMC electrodes.

organic electrolyte. The CV and CD measurements are conducted, and the results are shown in Fig. 4c and d respectively. The symmetric supercapacitor exhibits specific capacitance values of 230, 217, 192, 172, 154, 128 and $114 \, \mathrm{Fg}^{-1}$ respectively at scan rates 2, 5, 10, 20, 50, 100 and 200 mV s⁻¹. The specific capacitance values obtained in this study are lower than those reported for VO₂ electrodes in aqueous electrolytes. However, this can be readily explained owing to the higher viscosity and lower conductivity of organic electrolyte PC as compared to water. The specific capacitance values decrease rapidly at higher scan rates and stabilizes at lower scan rates. At lower scan rates and current densities, electrolyte ions in the organic electrolyte get enough time to diffuse from one electrode to another contributing to larger capacitance values.

The variation of capacitance with current density obtained from charge-discharge curves is shown in Fig. S2, which shows that a specific capacitance of about $170 \, \text{Fg}^{-1}$ is obtained for the symmetric capacitor at a constant current density of 1 Ag⁻¹. The supercapacitor retains a specific capacitance of $75 \, \text{Fg}^{-1}$ at a very high current density of 40 Ag⁻¹, indicating reasonably good transport properties of the device. The Ragone plot for the symmetric supercapacitor based on VO₂ electrodes and the liquid organic electrolyte is shown in Fig. 4e. The energy and power density values in the plot are derived from charge-discharge curves at different current densities. At a constant power density of $25\,kW\,kg^{-1},$ the supercapacitor exhibited an energy density of $35\,Wh\,kg^{-1}.$ At a low power density of $1.4\,kW\,kg^{-1},$ the energy density reached as high as 46 W h kg^{-1} , much higher than that reported for VO₂ based aqueous symmetric capacitors [54,56,58,61–64]. These results indicate that even though VO₂ based organic symmetric capacitor exhibits lower specific capacitance performance than the aqueous capacitor, the wide operating voltage range of the organic electrolyte system (1.4 V) contributes to the significant enhancement of the energy and power density performance. The long-term cycling stability of the VO₂ based symmetrical supercapacitor is tested using galvanostatic chargedischarge cycling at a constant current density of 10 Ag^{-1} . The variation of capacitance with cycle number is shown in Fig. 4f. A sharp increase in the specific capacitance (increase of nearly 6% of the initial capacitance) is observed in the initial 500 cycles, which can be explained as the activation process of the electrode materials. Long charging-discharging process may help electrolyte ions intercalate into the spaces between VO₂ nanosheets, fully utilizing the electroactive surface area of electrode materials. Due to this Li intercalation enhanced pseudocapacitance, the maximum

specific capacitance is achieved at 500th cycle. After that there is a gradual decrease in the capacitance values to next 800 cycles (loss of nearly 10% of the maximum capacitance). The presence of oxygen-containing hydrophilic functional groups and moisture on the surface of VO₂ can be responsible for this degradation in the specific capacitance in organic electrolyte. The specific capacitance becomes quite stable after that (slight decrease with number of cycles). The VO₂ based organic symmetric capacitor could retain nearly 82% of its maximum capacitance (more than 95% of the initial capacitance) even after 6000 charge-discharge cycles.

The charge transfer kinetics of symmetric supercapacitor based on VO₂ electrodes and the liquid organic electrolyte was investigated using the electrochemical impedance spectroscopy in the frequency range of 100 kHz-1 mHz, and the Nyquist plot is shown in Fig. 5a. The plot shows the semicircle in the highfrequency region and a straight line in the low-frequency region. The X intercept of the Nyquist plot in the high-frequency region gives the value uncompensated resistance of the bulk electrolyte solution (R_s) . The diameter of the semi-circle in the high-frequency region corresponding to the charge-transfer resistance (R_{ct}) . For the VO₂ based symmetric supercapacitor, The R_s and R_{ct} values obtained from the Nyquist plot are 2.6 and 0.9 Ω respectively. The low value for R_{ct} indicates high conductivity of the VO₂ composite electrodes with excellent electrolyte accessibility. The line at the low-frequency region, making an angle 45° with the real axis, is called the Warburg line resulting from the frequency dependence of ion diffusion in the electrolyte to the electrode interface. Variation of specific capacitance with frequency for symmetric supercapacitor based on VO₂/C/CMC electrodes is shown in Fig. 5b. The plot indicates a pseudocapacitive behavior with a continuous increase in the specific capacitance with the decrease in frequency.

Developing energy storage devices which can be integrated into the flexible electronic devices has attracted considerable attention. An essential requirement to develop solid-state flexible energy storage devices is to have a solid-like gel electrolyte which can facilitate ionic conduction in the solid-state device. Fumed silica has been used to form ion gels using ionic liquids[65]. Here, we have developed a slightly different approach to form the fumed silica gel. Instead of using expensive ionic liquids, the liquid electrolyte solution of 1 M LiClO₄ in propylene carbonate is used to form the gel with fumed silica. The 1 M LiClO₄ solution of propylene carbonate acts as a gelling agent as well as the electrolyte. The resultant fumed silica gel is quite stable and transparent. The CVs of the solid-state symmetrical device fabricated with CMC composite electrodes is shown in Fig. 6a.



Fig. 6. (a) CVs at different scan rates (b) Charge-discharge characteristics at different current densities and (c) variation of capacitance with current densities for VO₂/C/CMC symmetric supercapacitor device in gel electrolyte, and (d) CVs of solid state device measured at different angles (inset shows the schematic for the measurement of the bending angles and the optical image of the flexible supercapacitor device).

The CVs show almost rectangular shape in the gel electrolyte. The charge-discharge characteristics of the solid-state devices at different rates are shown in Fig. 6b. A specific capacitance of about $145 \, \text{Fg}^{-1}$ and a gravimetric cell capacitance of $36 \, \text{Fg}^{-1}$ are achieved at a discharge current density of $1 \, \text{Ag}^{-1}$. The decrease in capacitance could be due to the decrease in ion diffusion properties of the gel electrolytes when compared to the liquid electrolytes. The variation of specific capacitance and gravimetric cell capacitance with different current densities for the solid state device is

shown in Fig. 6c. At the high current density of 25 Ag⁻¹, the device retains a specific capacitance of 84 Fg⁻¹ and a cell capacitance 21 Fg⁻¹, demonstrating good transport phenomena required for the fast delivery of energy of the gel electrolyte. The flexibility of the solid state device was examined by conducting the CV measurements at different bending angles at a constant scan rate of 50 mV/s and the results are shown in Fig. 6d. The CVs show negligible change in currents at different bending angles compared to the flat configuration. The shape of the CVs at different bending



Fig. 7. Optical images displaying depth of discharge, lighting up the LED using three devices which are connected in series at different intervals of time (a) after 5 s, (b) after 20 s and (c) after 50 s.

angles remains almost the same, demonstrating that the solidstate device can be integrated into flexible platforms. Further, we have connected the three solid-state devices in series to power a red LED. The optical images taken at different time intervals of lighting an LED with the fully charged energy storage devices are shown in Fig. 7a–c. The solid-state device could power the LED for more than a minute.

4. Conclusions

2D VO₂ nanosheets with high surface area and excellent electrochemical properties were synthesized by the simultaneous exfoliation and reduction of V₂O₅ bulk powder by a hydrothermal route. Flexible supercapacitor electrodes were fabricated by drop-casting water-based suspension containing VO₂ nanosheets, conducting carbon and CMC binder over carbon paper. The symmetric supercapacitor based on VO₂ nanosheet electrodes and liquid organic electrolyte (1 M LiClO₄ in PC) exhibited an energy density of 46 Wh kg⁻¹ at a power density of 1.4 kW kg⁻¹ and a constant current density of 1 Ag⁻¹. Flexible solid-state symmetric supercapacitor was also fabricated using organic gel electrolyte and VO₂ nanosheet electrodes which exhibited a specific capacitance of 145 Fg⁻¹ and a device capacitance of 36 Fg⁻¹ at a discharge current density of 1 Ag⁻¹. The performance of the solid-state device was unaltered at different bending angles and series combination of three such devices could power an LED for more than a minute.

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Appendix A. Supplementary data

Supplementary data associated with this article can be found, in the online version, at http://dx.doi.org/10.1016/j. electacta.2016.10.109.

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